# DSC and TSC studies of polyurethane mono- and bilayer paints<sup>1</sup>

L. Drouet-Fleurizelle<sup>a,b</sup>, D. Gillereau<sup>b</sup> and C. Lacabanne<sup>a,\*</sup>

<sup>a</sup> Solid State Physics Laboratory (URA 74), Paul Sabatier University, 118 Route de Narbonne, 31062 Toulouse Cédex (France) <sup>b</sup> Aerospatiale, Aircraft Division, Quality Department, 316 Route de Bayonne, 31000 Toulouse (France)

(Received 30 November 1992; accepted 14 May 1993)

## Abstract

Differential scanning calorimetry and thermally stimulated current spectroscopy have been applied to the characterization of the microstructure of polyurethane/polyurethane bilayers. The microstructure of the constituting polyurethane monolayers was used as reference. It is constituted of two phases indicating the segregation of soft and hard sequences. In bilayers, the microstructure is strongly dependent upon the recovering time. The microstructure is also modified upon ageing.

# INTRODUCTION

In aeronautics, the external aircraft paint is constituted by a polymeric multilayer. The aim of this work is to characterize the outer polyurethane–polyurethane bilayer for various processing conditions. The phenomeno-logical characteristics of the bilayer are strongly dependent upon the drying time of the inner layer before application of the outer layer. Analysis of the molecular mobility of polyurethane gives a characterization of its microstructure [1-3].

In this work, two thermo-stimulated spectroscopies were used: differential scanning calorimetry (DSC) and thermo-stimulated current (TSC) spectroscopy [4]. A preliminary study of each polyurethane monolayer will be used as reference.

<sup>\*</sup> Corresponding author.

<sup>&</sup>lt;sup>1</sup> Presented at the 21st Annual NATAS Conference, Atlanta, GA, 13–16 September 1992.

## MATERIALS AND METHODS

Paint was deposited onto an aluminium support. The inner layer constituting the barrier coating (BC) was  $25 \,\mu$ m thick. Two recovering times (RT), 0.25 and 8 h, were used for application of the outer layer, i.e. the finition (F) which was  $80 \,\mu$ m thick. The conditions of ageing were constant at 23°C and 55% relative humidity.

For DSC experiments, flakes of paints were compressed in order to fill the aluminium pan. Experiments were carried out on a Perkin-Elmer DSC-7 with a heating rate of 20°C min<sup>-1</sup> in the temperature range -60to +250°C. For each sample, two consecutive runs were recorded. Only second-run thermograms are discussed here.

For TSC experiments, the metallic support of the bilayer was used as one electrode. An identical plate was used as the counter electrode. Experiments were carried out on a Solomat TSC-RMA spectrometer, with a heating rate of  $10^{\circ}$ C min<sup>-1</sup> in the temperature range from RT to +160°C. All the samples were polarized under the same conditions, i.e. 2 min at 100°C and 200 V, quenched at  $-30^{\circ}$ C.

DSC STUDY

## Monolayers

Figures 1 and 2 show the glass transition temperatures  $T_g$  measured on the DSC thermograms recorded as a function of ageing time, for the two polyurethane monolayers BC and F, respectively.



Fig. 1. Evolution of  $T_g$  as a function of ageing time for BC monolayer.



Fig. 2. Evolution of  $T_{g}$  as a function of ageing time for F monolayer.

BC is characterized by values of  $T_g$  that are high for a polyurethane material. The  $T_g$  values increase strongly from  $T_g = 54^{\circ}$ C to  $T_g = 77^{\circ}$ C, after seven weeks of ageing time. It is important to note here that rapid kinetics are involved in the polymerization process.

F is characterized by low values of  $T_g$  (11°C) after 2 days of ageing time, and 16°C after 12 weeks of ageing time. The evolution of  $T_g$  values is less important than for BC. This result is probably related to the very low kinetics of the polymerization.

## Bilayers

Figures 3 and 4 show the DSC thermograms recorded for the bilayers with recovering times RT = 0.25 and 8 h, respectively.

A single step is observed in all cases, independent of the recovering time or ageing time. The  $T_g$  values are intermediate between those of BC and F.

This phenomenological miscibility might be a result of the chemical nature of both monolayers.

The inflexion point of the step has been associated with the glass transition temperature  $T_g$ . Table 1 lists the  $T_g$  values recorded for RT = 0.25 h and RT = 8 h, as a function of ageing time. We have also indicated, as reference, the  $T_g$  values of the BC and F monolayers.

It is also interesting to note the variation in the heat capacity step at  $T_{g}$ ,



Fig. 3. DSC thermograms for BC/F bilayer with a recovering time RT of 0.25 h for various ageing times: ---, 2 days; ----, 7 weeks; ---, 12 weeks.



Fig. 4. DSC thermograms for BC/F bilayer with a recovering time RT of 8 h for various ageing times: —, 2 days; —, 7 weeks; ---, 12 weeks.

Ageing time	Bilayer		Monolayer		
	RT = 0.25 h	RT = 8 h	BC	F	
2 days	42.5	40.5	54	11	
7 weeks	46	41	76	13	
12 weeks	47	42	77	16	

TABLE 1

Variation of  $T_{\sigma}$  from DSC experiments as a function of ageing time

 $\Delta C_p$ .  $\Delta C_p$  is lower for RT = 0.25 h than for RT = 8 h. This evolution confirms the restriction of molecular mobility for the shorter recovering time.

# TSC STUDY

## Monolayers

Figures 5 and 6 show the TSC thermograms recorded for the two monolayers, BC and F, as a function of ageing time.

The TSC thermograms recorded for the BC coating show three relaxation peaks respectively situated at  $T_{m_i}$  (low),  $T_{m_i}$  (intermediate) and



Fig. 5. TSC thermograms for BC monolayer as a function of ageing time: —, 4 days;  $-\cdot -$ , 4 weeks; ---, 13 weeks.



Fig. 6. TSC thermograms for F monolayer as a function of ageing time: ---, 4 days; ---, 4 weeks; ---, 13 weeks.

 $T_{m_h}$  (high) in order of increasing temperature. The first peak is shifted from 81 to 121°C, with a decrease in intensity with ageing time. It probably corresponds to the dielectric relaxation of the soft sequences of the polyurethane. The second peak is shifted from 126 to 138°C and increases in intensity with ageing time. This relaxation mode is assigned to the molecular mobility of the hard polyurethane sequences that are strongly involved in the polymerization process of the inner coating BC. The third peak is probably the consequence of free charge movements.

The TSC thermograms recorded for the F outer layer also show three peaks. The first, associated with the dielectric relaxation of the soft sequences, increases in temperature and in intensity. It is interesting to note the influence of soft segments in the polymerization of the outer layer. All values measured on the different thermograms for the three peaks are lower than those measured for BC. The second peak increases in temperature, but decreases in intensity. It corresponds to the relaxation of the hard sequences. The third peak is characterized by a higher intensity than for BC. A more important mobility of the polyurethane sequences is involved in the polymerization process. It might facilitate charge transport.

#### Bilayers

Figures 7 and 8 show the TSC thermograms recorded for the bilayers with recovering times of 0.25 and 8 h, respectively.

For short recovering times, at 1 week of ageing, three relaxation peaks are observed. The lower temperature peak is situated at 70°C. It corresponds to the step recorded on the DSC thermograms and therefore it



Fig. 7. TSC thermograms for BC/F bilayers for a recovering time RT of 0.25 h.

has been attributed to the dielectric manifestation of the glass transition of the bilayer. The soft polyurethane sequences might be involved in this mode. The intermediate peak is situated at 100°C. According to previous data, this relaxation mode has been assigned to the molecular mobility of the hard sequences. The high temperature peak is located at 128°C. Free charge movements are probably involved in this process. Table 2 lists the  $T_m$ values corresponding to the maximum of the TSC peak for RT = 0.25 h and 8 h, as a function of ageing time.



Fig. 8. TSC thermograms for BC/F bilayers for a recovering time RT of 8 h.

Ageing time	Bilayer BC/F				Monolayer							
	RT = 0.25 h			RT = 8 h		BC		F				
	$\overline{T_{m_1}}$	T <sub>mi</sub>	T <sub>mh</sub>	$T_{m_1}$	T <sub>mi</sub>	T <sub>mh</sub>	$T_{m_1}$	T <sub>mi</sub>	$T_{m_h}$	$T_{m_1}$	$T_{m_i}$	$T_{m_h}$
4 days	70	100	127	75	100	134	81	126		50	80	120
4 weeks	95	112	133	91	108	139	97	123		55	85	129
13 weeks		108	140	107	120	142	121	138	160	72.5	95	131

TABLE 2			
Variation of $T_{m}$ from TSC experiments as	a function	of ageing	time

It is interesting to note that upon ageing, the contribution of the low temperature peak is strongly decreased [5]. This confirms the hypothesis proposed for explaining previous DSC data. For long recovering times, the intensity of the low temperature peak is smaller so that it is not well-resolved on the TSC spectrum for 1 week of ageing. It can only be distinguished after 3 weeks. This indicates a softening of this bilayer by comparison with the previous case.

## CONCLUSION

Differential scanning calorimetry and thermally stimulated current spectroscopy are well adapted to the investigation of molecular mobility in polyurethane coatings.

Charactersitic parameters of the monolayers are proposed, and the contributions of the hard and soft sequences have been identified.

The study of the bilayers has shown the existence of a phenomenological miscibility of the hard and soft sequences. This pseudo-miscibility is strongly dependent upon the recovering time.

This methodology is well adapted to the study of ageing procedures in polymeric coatings.

## REFERENCES

- 1 T. ElSayed, Thèse Ingénieur Docteur, University of Toulouse, 1979.
- 2 T. ElSayed, D. Chatain and C. Lacabanne, in G. Astarita, G. Marucci and L. Nicholais (Eds.), Rheology, Vol. III, Plenum Press, New York, 1980, pp. 171–175.
- 3 J.T. Koberstein, I. Gancarz and T.C. Clarke, Technical Report, Navy Department, 1989 pp. 1–19.
- 4 A. Bernès, R.F. Boyer, D. Chatain, C. Lacabanne and J.P. Ibar, in S.E. Keinath, R.L. Miller and J.K. Ricke (Eds.), Order in the Amorphous State of Polymers, Plenum Press, New York, 1986, pp. 305–326.
- 5 L. Drouet-Fleurizelle, Thesis, University of Toulouse, 1992.